

Basalt fibre degradation in seawater and consequences for long term composite reinforcement

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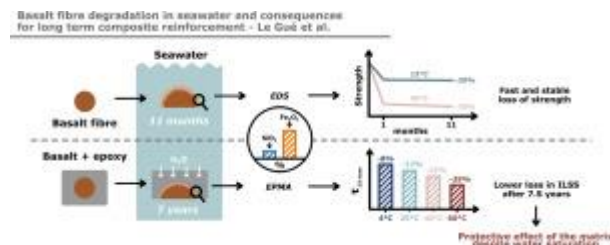
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Abstract :

Basalt fibres are increasingly employed as reinforcements in marine composites, but their behaviour in natural marine environments is underexplored. This study investigates basalt fibre ageing in renewed natural seawater at 15 °C and 40 °C. After one month in seawater at 15 °C and 40 °C, tensile strength dropped, stabilizing at approximately –40% and –60%, respectively. This rapid initial property decline, followed by slower degradation, is attributed to an altered surface layer on the fibres. Initially causing significant property loss, this layer then plays a protective role, preserving the fibre core. The impact on basalt/epoxy composites exposed to 7.5 years of seawater was less severe, with a 20% loss at 40 °C, demonstrating the protective function of the matrix. This study suggests that basalt fibres undergo rapid, then stable, property degradation in water, but remain suitable for use as epoxy matrix composite reinforcements, thanks to the protective role of the resin.

Graphical abstract



Highlights

► Basalt fibres were immersed in seawater for up to 11 months. ► Basalt/epoxy composites were exposed to seawater for 7.5 years. ► The fibres alone experienced rapid property loss, followed by stabilization. ► Immersed fibres displayed a degraded surface layer rich in iron oxide and low in silicon oxide. ► For composite samples the protective role of the epoxy matrix resulted in minimal strength loss.

Keywords : basalt fibres, composites, seawater ageing

29 **1. Introduction**

30 In recent years, the marine industry has been actively engaged in an ecological transition to reduce its
31 environmental impact. This shift involves adopting more environmentally-friendly practices, including the
32 exploration of alternative materials with lower ecological footprints compared to conventional ones.

33 Within this context, the utilization of basalt fibres as a substitute for glass fibres for composites is
34 gaining popularity in boatbuilding, despite the lack of comprehensive data on the life cycle inventory of
35 individual fibres in the scientific literature. However, as the properties of these fibres are inferior to those
36 of carbon fibres, they are not intended to replace them ([Chowdhury et al., 2022](#)). Current applications of
37 basalt fibres range from the reinforcement of polymers in the automotive industry ([Wang et al., 2021b](#)), to
38 the reinforcement of cement in civil engineering ([Sim et al., 2005](#)). Basalt fibres are produced through the
39 grinding and extrusion of volcanic rock resulting from the rapid cooling of magma in the ocean ([Militky](#)

40 [et al., 2002](#)). While the composition of basalt is similar to glass, it contains iron oxides in addition to silica,
 41 which affects the final properties of the fibres ([Fiore et al., 2015](#); [Austin and Subramanian, 1979](#)). However,
 42 the presence of iron in the fibres also leads to interactions between their surfaces and the surrounding
 43 environment as described by several authors ([Burkhard and Scherer, 2006](#); [Wei et al., 2011](#)). The marine
 44 environment is known for its challenging conditions, with high concentrations of ions ([Millero et al., 2008](#)),
 45 making basalt fibres susceptible to degradation. Table 1 presents a list of published studies on the ageing
 46 of basalt fibres or basalt composites in different aqueous environments.

Reference	sample type	ageing environment	ageing temp.	max ageing time	year
(Scheffler et al., 2009)	fibres	alkaline	20, 40, 60, 80°C	15 days	2009
(Wei et al., 2011)	composite	artificial seawater	25°C	3 months	2011
(Förster et al., 2014)	fibres	alkaline	80°C	11 days	2014
(Wu et al., 2015)	fibres	alkaline, acidic,	25, 55°C	66 days	2015
	+ composite	saltwater, tap water			
(Rybin et al., 2016)	composite	alkaline	"ambient"	64 days	2016
(Quagliarini et al., 2016)	composite	alkaline, acidic	100°C	3 hours	2016
(Davies and Verbooue, 2018)	composite	natural seawater	4, 25, 40, 60°C	200 days	2018
(Tang et al., 2018)	fibres	alkaline	25, 50, 70°C	3 days	2018
(Wang et al., 2019)	composite	artificial seawater	25°C	6 weeks	2019
(Lu et al., 2020)	composite	alkaline, tap water,	20°C	180 days	2020
		artificial seawater			
(Wang et al., 2021a)	fibres	artificial seawater	80, 85, 90°C	168 hours	2021
(Lu et al., 2022)	fibres	artificial seawater,	20, 40, 60°C	90 days	2022
	+ composite	distilled water			

Table 1: List of published studies on the ageing of basalt fibres or basalt composites in different aqueous environments.

47 Basalt fibres are often depicted as alternative fibres in civil engineering and cement composites ([Fiore](#)
 48 [et al., 2015](#); [Jiang et al., 2022](#)), which explains why the majority of ageing studies in the Table 1 refer to
 49 their durability in alkaline solutions ([Scheffler et al., 2009](#); [Förster et al., 2014](#); [Wu et al., 2015](#); [Rybin et al.,](#)
 50 [2016](#); [Quagliarini et al., 2016](#); [Tang et al., 2018](#)). Fibre degradation was observed after immersion in both
 51 NaOH solutions ([Scheffler et al., 2009](#); [Förster et al., 2014](#); [Wu et al., 2015](#); [Rybin et al., 2016](#); [Quagliarini](#)
 52 [et al., 2016](#); [Tang et al., 2018](#)) and CaOH solutions ([Quagliarini et al., 2016](#)). Some authors stated that the
 53 degradation was linked to the dissolution of the glass network during ageing ([Scheffler et al., 2009](#); [Förster](#)
 54 [et al., 2014](#)). This degradation could be slowed down by the formation of a corrosion shell at the surface
 55 of the fibre ([Förster et al., 2014](#); [Tang et al., 2018](#)) which is a temperature dependant process ([Tang et al.,](#)

56 2018). Fibres can also be protected inside a polymer matrix (Quagliarini et al., 2016), but the corrosion
57 of the surface could degrade the fiber-matrix interfaces, leading to a loss of mechanical properties of the
58 composite (Wu et al., 2015).

59 Saltwater and artificial seawater ageing has also been applied to basalt fibres (Wang et al., 2021a; Lu
60 et al., 2022) and their composites (Wei et al., 2011; Wang et al., 2019, 2021a; Lu et al., 2022). Wang et al.
61 (2021a) immersed basalt fibres in artificial seawater at high temperatures and first noticed a fast increase
62 in the mechanical properties followed by a decrease. The authors stated that the increase was due to the
63 smoothing of the micro-cracks present at the surface of the fibres due to the corrosion of the crack tips.
64 The decrease was attributed to the glass network degradation by corrosion. After immersing basalt fibres
65 in artificial seawater at lower temperatures, Lu et al. (2022) also observed an important decrease in the
66 mechanical properties, without giving further details on the degradation process. Different authors showed
67 that using a polymer matrix could protect the fibres from the degradation process (Quagliarini et al., 2016;
68 Lu et al., 2022) which explains the enhanced durability observed at the composite scale in other studies (Wei
69 et al., 2011; Davies and Verbooue, 2018; Wang et al., 2019). However, basalt fibre reinforced composites
70 are considered more sensitive to water uptake than glass fiber composites (Davies and Verbooue, 2018; Lu
71 et al., 2022). While some authors stated that the long term response of both composites are similar (Wei
72 et al., 2011; Davies and Verbooue, 2018), other observed a poorer resistance to seawater ageing for the
73 basalt composite (Lu et al., 2020) due to the formation of a corrosion layer that degrades the fiber-matrix
74 interface adhesion (Wang et al., 2019).

75 More generally this table highlights the increasing interest in basalt fibres in recent years, as most of the
76 studies listed are less than 10 years old. The only study conducted in natural seawater with a substantial
77 ageing period involves composites immersed for up to 200 days (Davies and Verbooue, 2018). Furthermore,
78 the other studies examining the ageing of basalt fibres alone in seawater were conducted using artificial
79 seawater (Wang et al., 2021a; Lu et al., 2022). The present study aims first to characterize the long-term
80 behaviour of basalt fibres in natural seawater at 15°C, close to the average temperature of the ocean, and at
81 40°C to facilitate the observation of degradation processes. These fibres are intended for integration into a
82 resin, leading to the testing of a basalt/epoxy composite after 7.5 years of immersion in seawater at different
83 temperatures to assess its very long-term ageing, and compare it to a reference glass/epoxy composite with
84 the same matrix.

85 2. Materials & methods

86 2.1. Materials

87 Two types of samples were studied here: continuous basalt fibres alone and two composites.

88 Continuous basalt yarns were supplied by Basaltex, a basalt fibres supplier based in Belgium. The
 89 physical properties of the yarn are presented in Table 2. No additives were added during the manufacturing
 90 process.

Density [$g.cm^{-3}$]	2.67
Tex [$g.km^{-1}$]	320.4
Diameter [μm]	10 – 15

Table 2: Physical properties of as received yarns.

91 Two different composite panels were supplied by Basaltex: one made of continuous basalt fibres, one
 92 made of continuous glass fibres, and both with the same epoxy resin. The dimensions of the panels were
 93 500x500 mm² with a thickness of 3 mm. The detailed constructions of both composites are presented in
 94 Table 3. The same matrix was used for both composites in order to compare the effect of the seawater ageing
 95 upon the different fibres. Panels were obtained by infusion by the Sirris company (Brussels, Belgium) and
 96 then post-cured 2 hours at 50°C and 2.5 hours at 80°C with ramps of 30 minutes between each step. These
 97 composites are the same as those studied by [Davies and Verbouwe \(2018\)](#).

	Fibres	Matrix
basalt/epoxy	Basaltex - BAS UNI 350 0° - 357 g.m ⁻² / 90° 50 g.m ⁻² + stitching 9 g.m ⁻²	Epoxy Araldite 1564 LY + Aradur 3687
E-glass/epoxy	Selcom - UNIE 300 0° - 300 g.m ⁻² / 90° 60 g.m ⁻² + stitching 13 g.m ⁻²	Epoxy Araldite 1564 LY + Aradur 3687

Table 3: Details on studied composites.

98 2.2. Ageing

99 The two types of samples, fibres alone and composites, were immersed in tanks filled with natural
 100 seawater. Yarn samples were wound around a polypropylene cylinder to avoid entangling during retrievals,
 101 and then placed inside a polypropylene pot. Composite panels were also placed in polypropylene pots for
 102 ageing. Plastic pots were then placed in the different ageing tanks. The seawater was pumped from Brest
 103 estuary and continuously renewed, providing a good representation of the conditions that the materials
 104 might experience in the marine environment. The tanks were maintained at various temperatures. The
 105 fibres alone were immersed in tanks regulated at 15°C and 40°C. The composite samples were immersed
 106 at 4°C, 25°C, 40°C and 60°C. Temperatures higher than those that materials may encounter in the marine

107 environment make it easier to observe certain physico-chemical degradation phenomena. Samples of fibres
 108 alone were removed after, 1, 3, and 7 months for tensile testing, while composite samples were removed for
 109 weighing over a period of seven and a half years, after which they were tested in interlaminar shear.

110 To gain a better understanding of the degradation of basalt fibres in an aqueous environment, ageing
 111 tests were also carried out using water deionized by osmosis, in order to assess the effect of ageing without
 112 specific ions present in seawater.

113 2.3. Mechanical testing

114 Yarns were tensile tested before and after ageing on an Instron™ 10 kN capacity test machine equipped
 115 with Instron™ pneumatic grips designed for yarn testing. The tests were controlled by setting a displacement
 116 speed of 50 mm/min. Strain was measured by following two markers with a Basler™ camera. All the tests
 117 were conducted in a room maintained at 21°C and 50% relative humidity. 20 samples were tested for the
 118 initial state, and 3 samples were tested for each ageing condition.

119 Composites were tested according to the ASTM D2344 standard to evaluate their interlaminar shear
 120 strength (ILSS). Samples were tested after 7.5 years of ageing in a saturated state, and their inter-laminar
 121 shear strengths were calculated according to Equation 1.

$$\tau_{13} = \frac{3 F_{max}}{4 B * d} \quad (1)$$

122 Where τ_{13} is the inter-laminar shear stress at failure, F_{max} is the maximum load measured during
 123 testing, B is the width, and d the thickness of the sample.

124 2.4. Surface characterization

125 Samples were observed by Scanning Electron Microscopy (SEM) using FEI Quanta 200 equipment. They
 126 were coated with a 60% gold / 40% palladium coating.

127 2.5. Composition analysis

128 Energy Dispersive Spectroscopy (EDS) has been used alongside SEM observations for qualitative inter-
 129 pretation of the observations. For a quantitative analysis, a CAMECA SX100 Electron Probe Microanalyser
 130 (EPMA) has been used on polished cross-sections (polished down to 0.25 micron grain size) coated with a
 131 carbon coating.

132 3. Results

133 3.1. Yarn

134 3.1.1. Initial state characterization

135 Twenty samples of the basalt yarn were tested, Figure 1 shows the stress-strain curves obtained. With
 136 a mean strain at break of 1.7%, a mean stress at break of 1457 MPa, and a mean modulus of 89.8 GPa, the

137 fibres exhibit a brittle behaviour common for mineral fibres. Table 4 summarizes the properties obtained
 138 from the test on the basalt fibres, together with properties for E-glass fibres taken from the literature (Xing
 139 et al., 2019). Basalt fibres and glass fibres have equivalent properties, with basalt fibres being slightly stiffer
 140 and E-glass fibres having a higher stress at break.

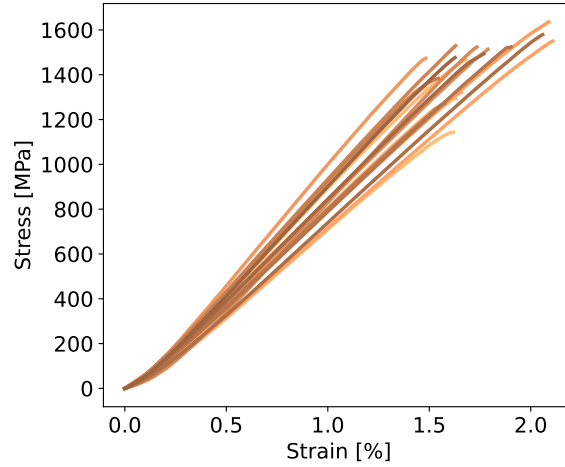


Figure 1: Initial tensile stress curves for basalt yarns.

	E [GPa]	$\epsilon_{failure}$ [%]	$\sigma_{failure}$ [MPa]
Basalt	77 – 106	1.5 – 2.1	1164 – 1669
E-glass	65.9 – 75.1	1.24 – 1.64	1280 – 2120

Table 4: Basalt and E-glass fibre properties (Xing et al., 2019).

141 3.1.2. Aged yarn sample testing

142 Basalt fibres samples were aged in tanks and retrieved after 1 month, 3 months, and 7 months. Three
 143 samples were tested for each condition without drying. Figure 2 shows the change in strength for samples
 144 aged at 15°C in seawater, 40°C in seawater, and 40°C in deionized water. The error bars represent the
 145 dispersion for each condition. After one month in 15°C seawater, basalt fibres have lost nearly 40% of their
 146 initial strength. This loss then stabilizes at a strength 45% lower after 7 months. The loss after one month
 147 at 40°C in seawater is more significant with a decrease in the strength of 86% and a stabilization around
 148 70% of the unaged value. In deionized water the loss is equivalent to that in seawater with a stabilization
 149 around 70% reduction.

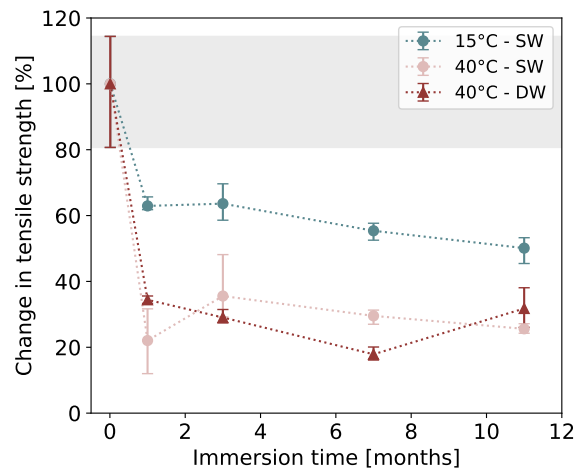


Figure 2: Change in strength for basalt yarn after different immersion times.

Immersion time [months]	1	3	7	11
15°C - seawater	-37%	-36%	-45%	-50%
40°C - seawater	-86%	-64%	-70%	-74%
40°C - deionized water	-66%	-82%	-71%	-68%

Table 5: Loss in strength for basalt yarns after different immersion times.

150 3.1.3. Surface characterization

151 After ageing, a change in the colour of the basalt fibre samples was observed which is shown in Figure 3.
 152 The filament in its initial state is a very dark brown, almost black colour. After two years in seawater,
 153 the filament lightened and lost its lustre. These observations have been made previously on basalt fibre
 154 composites aged in seawater and alkaline solutions (Wei et al., 2011; Wu et al., 2015).

155 The surfaces of the samples have also been analysed by SEM both in the initial state and after 7 months'
 156 immersion. Examples of these observations are presented on Figure 4. Figure 4 a shows the initial surface of
 157 the basalt fibres before ageing. The surface was smooth and clean with no defects. After 7 months at 15°C
 158 in seawater, the surface of the samples has changed and blisters have appeared. These were aligned in the
 159 fibre direction, and present on all fibres. Fibres immersed at 40°C in seawater during 7 months show holes,
 160 debonds, and crystals have formed on their surface as depicted in Figure 4 c. Different crystal patterns were
 161 observed. For fibres aged in 40°C deionized water, presented on Figure 4 d, the observations were the same
 162 as for the fibres immersed at the same temperature in seawater with crystals and debonds.

163 On Figure 4 c and d the shaded area represents one of the debonds observed during the analysis, it
 164 is then possible to see a degraded layer with a thickness of less than 1 micron. Several authors observed

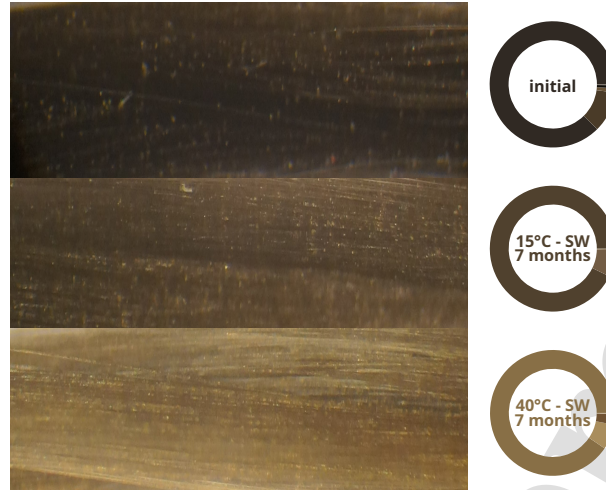


Figure 3: Change in colour observed for the basalt yarns after 7 months in seawater.

165 similar crystals formation and debonded layers on basalt fibres aged in alkaline solution (Scheffler et al.,
 166 2009; Rybin et al., 2016; Tang et al., 2018).

167 3.1.4. Composition analysis

168 EPMA has been used on 10 different fibres to characterize the composition of the fibre at the initial
 169 state. The initial composition obtained is summarized in Table 6 below.

	P ₂ O ₅	Na ₂ O	MgO	SiO ₂	Al ₂ O ₃	K ₂ O	CaO	TiO ₂	Fe ₂ O ₃	Cr ₂ O ₃	MnO	NiO
initial	0.3	2.0	3.9	54.3	17.4	1.7	8.2	1.1	9.8	0.005	0.2	0.002

Table 6: Initial composition of basalt fibre.

170 EDS has also been utilized in conjunction with SEM observations, in order to observe alterations in
 171 morphology. Figure 5 presents the results of two composition analyses conducted on fibres aged for 7 months
 172 at 40°C in seawater, with the initial composition included for comparison purposes. Crystals observed in
 173 Figure 4 (c) were analysed and labelled as "A", while a crystal-free surface was also analysed and marked as
 174 "B." In both areas, a significant decrease in SiO₂ content was observed, with less than 10% remaining on the
 175 surface lacking crystallization. Furthermore, an enrichment of Fe₂O₃, CaO, and TiO₂ was detected. The
 176 changes observed by EDS were more pronounced in the zone without crystalline structures. It is important to
 177 note that the analyses were performed on non-flat surfaces, which cannot therefore be used for quantitative
 178 comparisons. However, the observed magnitude of changes allows qualitative assessments to be made.

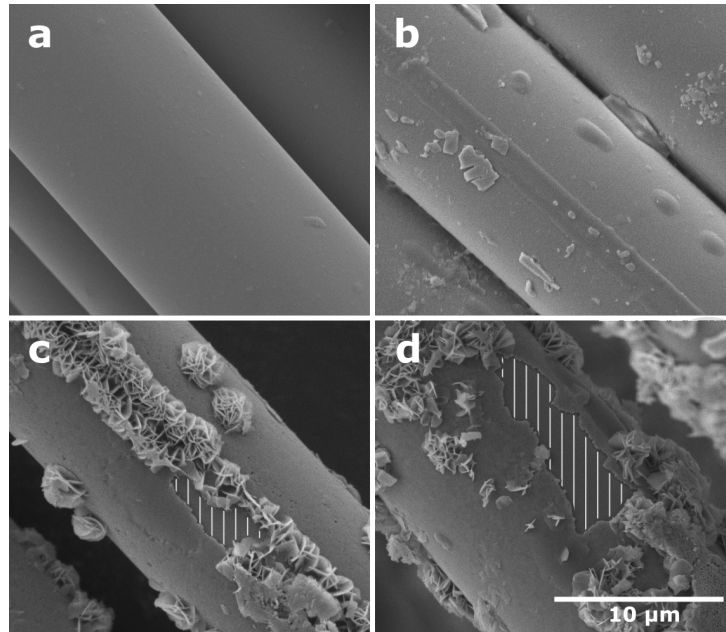


Figure 4: SEM observations of basalt fibres : a) at the initial state, b) after 7 months of immersion in a 15°C seawater, c) after 7 months of immersion in a 40°C seawater, d) after 7 months of immersion in a 40°C deionized water.

179 3.2. Composites

180 3.2.1. Water uptake

181 The laminates described in Table 3 were immersed in seawater at various temperatures: 4°C, 25°C,
 182 40°C, and 60°C. The mass gain of three samples was regularly monitored for one year, and then observed
 183 occasionally over a longer period as the samples were immersed for 7.5 years.

184 Figure 6 shows the composite samples when they were removed from water after 7.5 years' immersion.
 185 There is a clear colour change, particularly marked at 60°C, for composites with both fibres. After 7.5 years
 186 in 40°C seawater, the basalt fibres appear to be lighter and orange at certain locations. After 7.5 years at
 187 25°C, basalt fibres inside basalt/epoxy samples were looking slightly lighter. No change in the colour were
 188 observed for samples immersed at 4°C.

189 Figure 7 illustrates the different water absorption curves for each temperature. Table 7 presents the
 190 diffusion coefficients obtained by Davies and Verbouwe (2018) and updated mass at saturation. As the
 191 temperature rises, the diffusion coefficient shows an increase. At temperatures below 60°C, the mass at
 192 saturation ranges from 1.22% to 1.55% for both materials. However, at 60°C, saturation is not reached for
 193 either material, and the water content in the material increases rapidly.

194 One sample of basalt/epoxy composite aged at each temperature was then dried in an oven at 40°C.
 195 Figure 8 shows the desorption plots. The desorption rates are equivalent for both materials and across the
 196 different immersion temperatures. No loss of material was observed after 4 months of drying at 40°C.

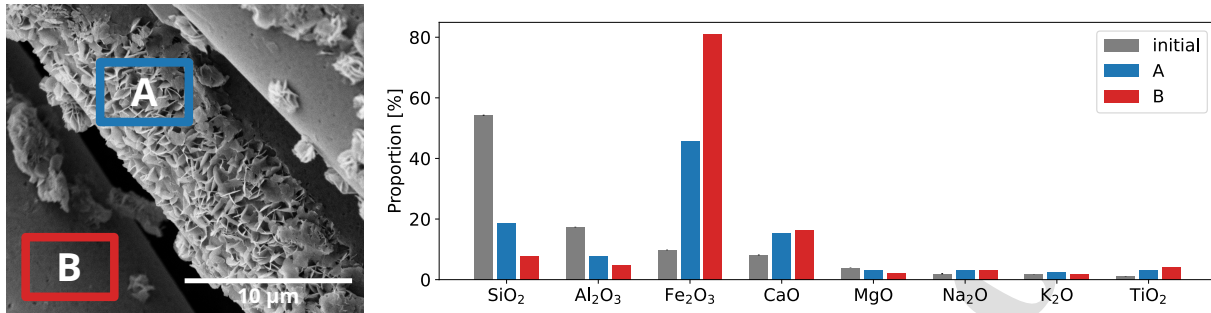


Figure 5: EDS analysis of basalt fibre surface after 7 months in seawater.

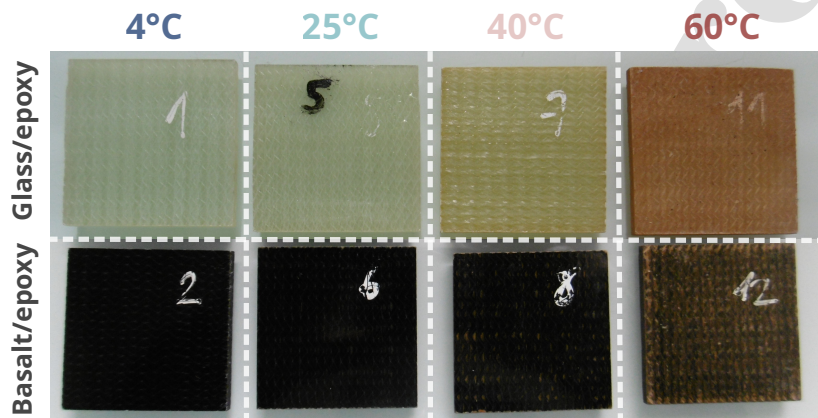


Figure 6: Weight gain samples after 7.5 years' immersion at 4 temperatures.

197 3.2.2. ILSS tests

198 The basalt/epoxy and glass/epoxy composites used for weight gain measurements were cut into inter-
 199 laminar shear test specimens following the ASTM D2344 standard and tested to failure. The first tests
 200 were performed on samples in their initial state and then after saturation in seawater for 200 days at 40°C
 201 (Davies and Verbouwe, 2018). The results obtained for these two laminates are presented in Table 8.

202 In the initial state, the glass/epoxy composite exhibited higher interlaminar shear strength, with a
 203 failure stress of 48.0 MPa, while the basalt composite failed at 44.1 MPa. After saturation, both laminates
 204 experienced approximately a 20% reduction in strength.

205 3.2.3. Long term seawater ageing

206 Basalt/epoxy and E-glass/epoxy samples were retrieved after 7.5 years of ageing in seawater at various
 207 temperatures. To investigate the influence of ageing after saturation, these samples were subjected to the
 208 same testing method as the initial interlaminar shear strength (ILSS) tests. Figure 9 presents the results
 209 for the maximum τ_{13} obtained. The bars labelled as "initial" represent the ILSS values obtained in the
 210 saturated state after 200 days at 40°C.

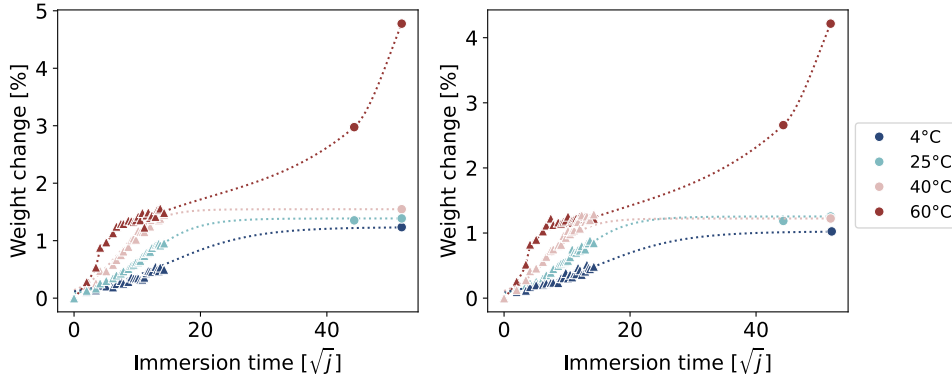


Figure 7: Weight change at different temperature for a) basalt/epoxy, and b) E-glass/epoxy.

(▲): (Davies and Verbouwe, 2018), (●): present study

$m^2/s \times 10^{-12}$	4°C	25°C	40°C	60°C
Basalt/epoxy	2.4 (0.3)	7.0 (0.8)	19 (2)	79 (5)
	1.24%	1.39%	1.55%	4.78%
E-glass/epoxy	1.8 (0.8)	7.5 (0.7)	21 (1)	96 (8)
	1.24%	1.25%	1.22%	4.22%

Table 7: Weight change at different temperature for basalt/epoxy and E-glass/epoxy

211 At 4°C, the ILSS of the basalt/epoxy composite exhibited a -7% reduction, while the E-glass/epoxy
 212 composite showed no significant change in strength, experiencing a -2% loss. When immersed in seawater
 213 at 25°C, the ILSS of the basalt/epoxy composite decreased by -25%, whereas the E-glass/epoxy composite
 214 maintained its strength with a non-significant loss of -2%. As the temperature increased to 40°C, the reduc-
 215 tion in ILSS became more pronounced. The basalt/epoxy composite experienced a -25% decrease, while the
 216 E-glass/epoxy composite also displayed a slight loss in strength, decreasing by -8%. At the highest temper-
 217 ature of 60°C, both composites demonstrated similar reductions in ILSS, with the basalt/epoxy composite
 218 showing a larger decrease of -39% compared to the -30% loss observed in the glass/epoxy composite.

219 3.2.4. SEM observations

220 Polished cross-sections of basalt/epoxy composite samples were prepared to examine potential morpho-
 221 logical changes occurring due to prolonged contact between the fibre and seawater-saturated resin. Figure 10
 222 displays the observations made for a sample aged for 7.5 years at 40°C, specifically focusing on the com-
 223 posite/seawater interface. The balanced nature of the composite allowed us to observe both normal and
 224 longitudinal sections.

225 A degradation gradient of the fibres is observable between the outer region of the composite in contact

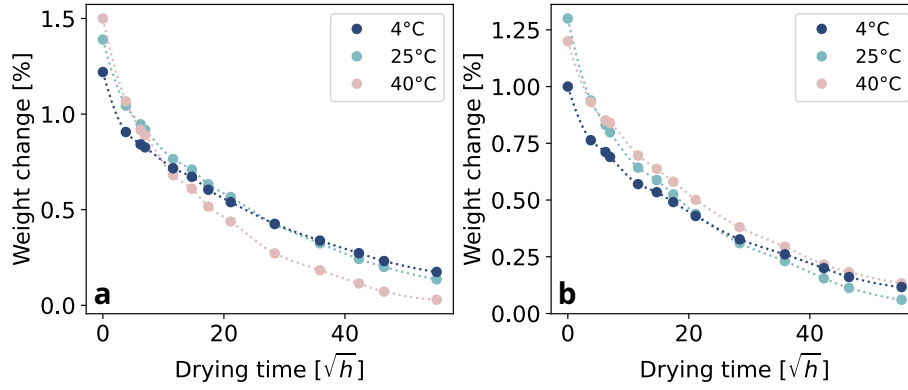


Figure 8: Water desorption after 7.5 years' immersion at 4 temperatures for a) basalt/epoxy, and b) E-glass/epoxy.

	τ_{13} [MPa]	τ_{13} saturated [MPa]	$\Delta\tau_{13}$
Basalt/epoxy	44.1 (1.7)	34.3 (0.7)	-22%
E-glass/epoxy	48.0 (1.6)	38.7 (0.7)	-19%

Table 8: τ_{13} for dry samples and samples saturated after 200 days at 40°C in renewed seawater according [Davies and Verbooue \(2018\)](#).

226 with seawater and the inner region within the first 30 microns. This degradation gradient is more significant
 227 for the fibres exposed at a 90° angle, and degradation effects are visible up to 80 microns into the composite.
 228 At the lower left of Figure 10 (a), a crack between two degraded fibres can be observed. The altered fibres
 229 consist of two distinct parts: a heavily degraded outer layer and an apparently intact core. This surface
 230 degradation has been observed previously by several authors after immersion in an alkaline medium ([Förster
 231 et al., 2014](#); [Scheffler et al., 2009](#)). At certain locations, the degraded layer appears to have been leached or
 232 washed away by seawater.

233 3.2.5. EPMA

234 EPMA analyses were conducted on the visible degradation features after 7.5 years of ageing in seawater
 235 at 40°C. Both the intact core of the fibre and the altered layer were examined to study the composition
 236 evolution with degradation and to compare the results with those obtained from individual fibres. Table 9
 237 presents the compositions at the fibre core and in the outer degraded layer.

238 The analysis of the fibre core shows a composition similar to the initial composition. However, significant
 239 changes are observed for the altered layer. For instance, the silica content decreases to 1.50% of the altered
 240 composition, while the iron oxide content increases and represents almost 30% of the total composition. The
 241 titanium oxide content also enriches, increasing from 1.11% to 3.83%. Contrary to the EDS observations,

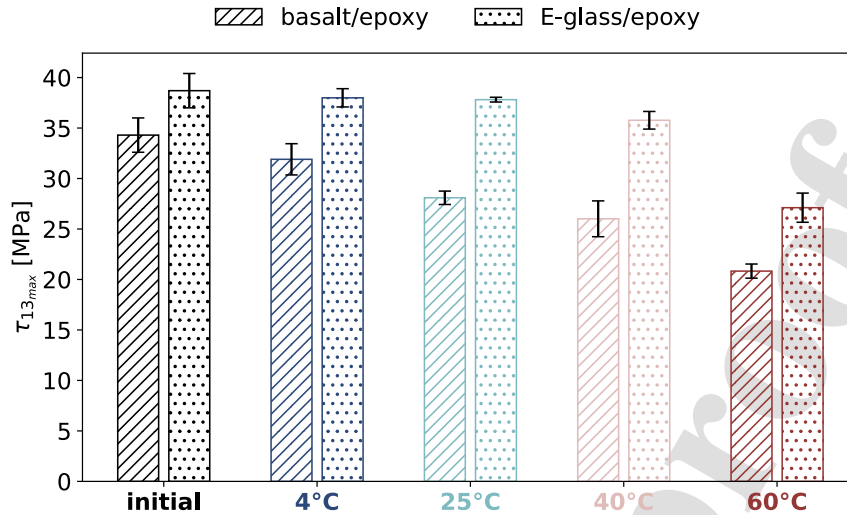


Figure 9: Mean residual τ_{13} after 7.5 year of ageing in renewed seawater at different temperatures.

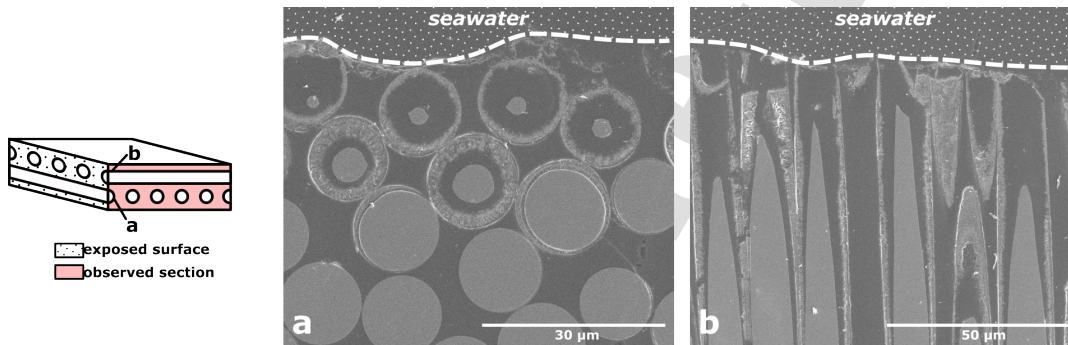


Figure 10: Cross (a) and longitudinal (b) sections of basalt fibres inside the epoxy matrix after 7.5 years of ageing in seawater at 40°C.

242 calcium oxide is no longer represented in the degraded part. Additionally, a substantial rise in magnesium
 243 oxide is observed, which was not evident in EDS results. Regarding silica, iron oxide, and titanium oxide,
 244 the EPMA findings align with the EDS results, providing consistent and complementary information.

245 4. Discussion

246 4.1. Fibre yarn degradation

247 Twenty samples of basalt fibres were tested to characterize the material in its initial state. The resulting
 248 strain-stress curves from these tests are shown in Figure 1, while the mechanical properties obtained for
 249 these basalt fibres are presented in Table 4. Values from the literature are provided in Table 10 below for
 250 comparison and discussion of the results.

	P ₂ O ₅	Na ₂ O	MgO	SiO ₂	Al ₂ O ₃	K ₂ O	CaO	TiO ₂	Fe ₂ O ₃	Cr ₂ O ₃	MnO	NiO
core	0.30	2.1	4.1	53.3	17.4	1.62	8.1	1.11	9.9	0.02	0.2	0
layer	0.37	0.13	21.7	1.50	11.6	0.02	0.48	3.83	28.7	0	0.06	0

Table 9: Fibre composition in the core and in the degraded outer layer after 7.5 years at 40°C [%].

251 The average Young's modulus obtained here, at 90 GPa, correlates well with values reported by various
 252 authors, which are also close to 90 GPa. However, for the properties at failure, the comparison with literature
 253 values is not as straightforward as for stiffness.

254 Tensile strength and elongation at break vary significantly among publications, with some authors re-
 255 porting tensile strengths almost five times higher than others. This wide variation can be attributed to the
 256 difficulty in testing this fragile material, which is sensitive to defects and testing conditions. This could
 257 explain why the Young's moduli are similar, but the properties at failure differ.

258 Additionally, the strength of basalt fibres is influenced by their thermal history (Sabet, 2015), and thus,
 259 their manufacturing process plays a crucial role. Furthermore, the composition of basalt also affects its
 260 mechanical properties (Austin and Subramanian, 1979). Currently, there is no robust classification of basalt
 261 fibres similar to that used for glass fibres.

E [GPa]	$\epsilon_{failure}$ [%]	$\sigma_{failure}$ [MPa]	Reference
		1846	Sabet (2015)
76	2.56	992	Sim et al. (2005)
76	1.8 - 3.2	1400 - 2500	Fiore et al. (2011)
80 - 90		1350 - 4750	Chairman and Kumaresh Babu (2013)
		1371 - 1489	Militky and Kovacic (1996)
90	3.15	4800	Lopresto et al. (2011)
93 - 110	3.1 - 6.0	3000 - 4840	Militký et al. (2018)
57.3	3.93	2250	Lu et al. (2022)

Table 10: Mechanical properties of basalt fibres published in the literature.

262 Basalt fibres being sensitive to their surrounding environment, ageing tests were conducted in seawater
 263 at 15°C and 40°C, and in deionized water at 40°C. Figure 2 presents the change in tensile strength obtained
 264 over a 7-month monitoring period.

265 Basalt fibres experienced a significant drop in their strength within the first month of immersion in
 266 seawater. This sudden loss of properties was followed by a stabilization at a plateau, which is dependent
 267 on the temperature but not on the environment, as observed for the ageing at 40°C. Similar stabilization
 268 at a comparable level has been reported by Wu et al. (2015) for ageing in both saltwater and tap water.

269 Several authors suggested a passive degradation layer forms and acts as a protective barrier slowing down
270 the degradation process of the fibres (Techer et al., 2001; Förster et al., 2014). This would explain the
271 plateau in strength loss. Wei et al. (2011) attributed the degradation of basalt to the oxidation of iron
272 present in the fibres by chloride complexes formed from Cl^- ions present in saltwater. However, in the
273 present study, the plateau level is the same after ageing in natural seawater rich in ions and ageing in
274 ultra-pure water where no ions are present. This suggests that the presence of Cl^- ions in the environment
275 does not significantly impact the ageing rate nor the plateau reached. Therefore, the ageing of basalt fibres
276 is governed by mechanisms other than the presence of chloride ions in the environment.

277 In addition to the loss of mechanical properties, changes in colour and a loss of lustre have been observed,
278 as shown in Figure 3. To gain a better understanding of the surface mechanisms at play in the fibres, SEM
279 analyses were conducted. The observations from these analyses after 7 months of immersion in the three
280 environments are presented in Figure 4. After 7 months at 15°C, the observed surfaces displayed aligned
281 bubbles or blistering, which can be considered as precursors of corrosion, potentially explaining the observed
282 loss of properties at this temperature. After 7 months of immersion in both seawater and deionized water,
283 the surface is altered, with the presence of debonds, holes, and crystal development. This kind of surface
284 modification has been observed previously by several authors (Rybin et al., 2016; Scheffler et al., 2009; Tang
285 et al., 2018; Wang et al., 2021a). The peeling of the corrosion layer does not seem to affect its protective role
286 regarding the stability of the mechanical properties after 11 months of immersion at 40°C for both seawater
287 and deionized water. Debonds are either negligible because they are very localised, or a new protective layer
288 of oxidation forms very quickly following debonding. EDS analysis was used to investigate the degradation
289 products present at the surface of the fibres. Both crystals and altered layers have been analysed, and the
290 results are presented on Figure 5. The large proportion of iron oxides on the outside of the fibre could be
291 linked to the oxidation of iron ions after glass network dissolution as observed by (Förster et al., 2014),
292 but could also be explained by a migration of iron ions outside the glass network as described by Michelin
293 et al. (2013). Similar iron oxide crystal patterns were observed by (Li et al., 2006). The corrosion shell
294 was detected despite the absence of crystals on the surface, exhibiting an even higher iron content. This
295 observation highlights the possibility that the layer could envelop the entire fibre, thereby confirming the
296 hypothesis that a protective corrosion layer has indeed developed. But while this layer could protect the
297 fibre from further degradation, it could also degrade the fibre-matrix interface when the fibre is used to
298 reinforce a composite.

299 4.2. Basalt composite durability

300 Basalt/epoxy and glass/epoxy composites have been immersed in seawater for 7.5 years at different
301 temperatures and weighed over time. The slopes of the weight gain curves on Figure 7 demonstrate that
302 the saturation rate increases with the immersion temperature as expected and that the saturation plateau

303 remains nearly the same for each temperature below 60°C. At that temperature, a similar initial plateau is
304 first observed, but followed by a continuous increase in water absorption. The glass transition temperature
305 of the dry resin ranges between 75°C and 80°C (Davies and Verbouwe, 2018). Water saturation is known to
306 decrease this temperature by approximately 20°C for epoxy resins due to plasticization (Le Guen-Geffroy
307 et al., 2019). With an ageing temperature of 60°C, which is close to or above the glass transition temperature
308 of the wet resin, prolonged immersion at this temperature could potentially result in chemical degradation
309 of the matrix and yellowing, as evidenced in Figure 6 (Antoon and Koenig, 1981; Krauklis and Echtermeyer,
310 2018). The continuous water uptake by both composites regardless of the fibre could be explained by an
311 increase in cracks and defects within the matrix, allowing water to penetrate more easily. This degradation
312 led to an important loss of mechanical properties for both composites as seen on ILSS tests after 7.5 years
313 of ageing at this temperature and makes it impossible to interpret the results regarding the degradation
314 of the fibres themselves. The absence of matrix degradation at lower temperatures suggests that these
315 conditions are more pertinent for assessing potential changes in the fibers or their interfaces. The discussion
316 will therefore focus on the temperatures below 60°C. For these temperatures, the fact that the weight change
317 plateaus are similar confirms the hypothesis that the mass at saturation can be described as independent of
318 the immersion temperature, even at 4°C. Furthermore, the water absorption appears to be almost completely
319 reversible for all temperatures, after 4 months of drying at 40°C, no loss of material is apparent. The observed
320 color change in the samples at both 25°C and 40°C, initially attributed to the presence of iron oxides based
321 on fibre results, appears to come from a very thin oxidized layer. This thin oxidized layer, while showing
322 minimal impact on weight gain based on desorption curves, seems to affect the mechanical properties of the
323 composites. After 7.5 years of immersion in natural seawater at 25°C and 40°C, the basalt/epoxy composites
324 experienced respectively 18% and 25% reductions in their interlaminar shear strength. This decline indicates
325 a loss in adhesion between the fibre and matrix at both temperatures, suggesting that the presence of the
326 oxidized layer is responsible for degrading the fibre-matrix interface. This assumption is supported by the
327 results obtained on the E-glass/epoxy composites where only a slight decrease of -8% was observed after 7.5
328 years at 40°C, which could be explained by the hydrolysis of the bond at the interface as observed by Devine
329 et al. (2023). However, the decline in the mechanical properties observed on basalt/epoxy composites is small
330 in comparison with the loss observed on the fibre alone, suggesting a protective role of the matrix as noted by
331 other authors (Wu et al., 2015; Quagliarini et al., 2016; Lu et al., 2022). The matrix protective role has been
332 investigated by SEM observations of the composite section that was in contact with seawater with results
333 presented on Figure 10. Heavily degraded fibres were then observed at the seawater/composite interface,
334 EPMA analysis revealed similar degradation products rich in iron oxides with almost no presence of silica,
335 suggesting that the basalt alteration was similar in both experiments. Furthermore, the presence of a fibre
336 degradation gradient indicates that the matrix has slowed down the diffusion of water within the composite,
337 thus slowing down the degradation process observed on fibres in direct contact with seawater. However,

338 water diffusion seems to be faster in the degraded regions and in cracks, which leads to an acceleration of the
339 degradation process as suggested by the degradation of the fibre near the crack seen on Figure 10 (a), and
340 on the longitudinal degradation observed on all fibres on Figure 10 (b). Corrosion at the surface of the fibres
341 not only reduces the fatigue life of basalt composites (Wang et al., 2019), but also accelerates degradation
342 by allowing water to enter via the degraded layer or cracks. Corrosion on the surface of the fibres has a dual
343 impact: it reduces the fatigue life of basalt composites, as discussed by Wang et al. (2019), and accelerates
344 degradation by enabling water diffusion through deteriorated layers or cracks. These interlinked processes
345 can combine, causing a quicker decline in mechanical properties under real-use conditions compared to
346 samples aged without mechanical stress.

347 5. Conclusion

348 This paper describes a study on the long term behaviour of basalt fibres and their epoxy reinforced
349 composites in natural seawater. The reinforcement alone was studied first and a fibre surface degradation
350 mechanism was identified. This was shown to be accompanied by the development of iron oxide (Fe_2O_3)
351 degraded layer during seawater ageing and a significant loss in fibre tensile properties. The degradation
352 in mechanical properties being significant even at low temperatures, it must be considered for applications
353 where basalt fibres are used uncoated and immersed, such as rope for fishing gear.

354 However, this loss is not directly transferred to reinforced epoxy composites, for which the reduction in
355 ILSS properties after 7.5 years in 40°C seawater is only around 20%. The polymer matrix provides long
356 term protection to the fibres which show a long term stability in composite form. These findings illustrate
357 the advantages gained by isolating the fibers from seawater using the matrix. However, they also highlight
358 the necessity of examining degradation of basalt/epoxy composites under realistic operational conditions.

359 CRediT authorship contribution statement

360 Louis Le Gué: Conceptualization, data gathering and investigation, formal analysis, visualization, writing
361 – original draft, writing - review and editing.

362 Peter Davies: Conceptualization, data gathering and investigation, formal analysis, writing – original
363 draft, writing - review and editing.

364 Mael Arhant: Conceptualization, investigation, formal analysis, writing - review and editing.

365 Benoit Vincent: Conceptualization, investigation, writing - review and editing.

366 Wouter Verbouwe: Conceptualization, investigation, writing - review and editing.

367 **Declaration of competing interest**

368 The authors received materials from Basaltex, and Wouter Verbouwe is an employee of Basaltex. Nev-
 369 ertheless, these affiliations did not influence the design, interpretation, or reporting of the findings.

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Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

Louis LE GUE reports equipment, drugs, or supplies was provided by Basaltex. Wouter Verbouwe reports a relationship with Basaltex that includes: employment.